| The second secon | Marie Control of the |
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| ACCESSION NR: AP4041587 | S/0078/64/009/007/1662/1668 |
| AUTHOR: Ko, Chih-ming; Kornilov, I | . I.; Py*layeva, Ye. N. |
| TITLE: Phase diagram of titanium-a | luminum-molybdenum-vanadium system |
| | imii, v. 9, no. 7, 1964, 1662-1668 |
| TOPIC TAGS: titanium aluminum allo vanadium containing alloy, alloy ph alloy property | www.molvbdenum containing alloy. |
| ABSTRACT: Sixty-nine alloys of the an A1 + Mo + V content of up to 50% mosphere of purified helium from iconum, 99.9% pure molybdenum, and 99% by microscopic, x-ray diffraction, by measurement of the hardness and were investigated in the as-cast coment. Isothermal sections of the on the basis of the microscopic and following phases and phase regions | were levitation merted in an acodide titanium, 99.99% pure alumi3% pure vanadium, and studied and dilatometric analysis, and electrical resistivity. Alloys ondition and also after heat treat- Ti-Al-(Mo:V = 1:1) system, plotted diversy phase analyses, showed the |
| Market and account to the control of | |
| | |

ACCESSION NR: AP4041587

建筑中央 医黑线性和 经国际部分的 人名英格兰人姓氏克尔特 医多种溶液 经未分配 医多种性神经病 化二氯

α+β, α+γ, β+γ, and α+β+γ. The data of the microscopic, x-ray, and dilatometric examinations were used to plot the polythermal sections of the system through the titanium corner and at Al to (Mo + V) weight-concentration ratios of 0:100, 15:85, 25:75, 50:50, 75:25, and 85:15. Examination of the plotted composition-hardness and composition-electrical resistivity diagrams showed that the hardness and electrical resistivity of most alloys increase with increasing summary concentration of Al, Mo, and V. The data on the electrical resistivity and hardness of alloys, quenched from 600C, also show the existence of the Ti₃Al compound in the Ti-Al-Mo-V system. Orig. art. has: 7 figures and 1 table.

ASSOCIATION: none

SUBMITTED: 07May63

ATD PRESS: 3067

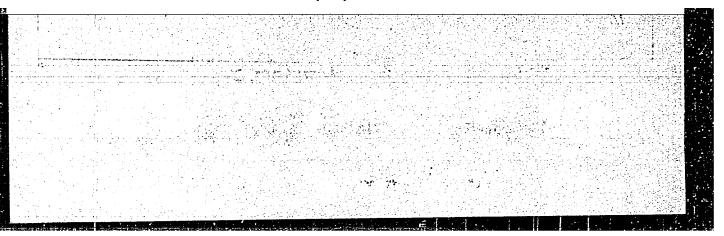
ENCL: 00

SUB CODE: MM

NO REF SOV: 006

OTHER: 000

Card 2/2



met?

J of the I ust of Hetals Feb. 1161 Properties of Alloys

Separation of Nickel Tantalide, Ni, Ta. from Alloys of the Binary System Nickel-Tantalum. I. I. Kornilov and E. N. Pylaeva (Dollady Mad. Nauk S.S.R. II. 1953, 91, (4), 841–8431.—(In Russian). By heating in cacuo at 1300°C, for 4 hr., at 1200°C, for 2 hr., at 1000°C, for 2 hr., and then slowly cooling to room temp., K. and P. produced a considerable coarsening of the Ni, Ta phase (present as elongated crystals) in the two-phase alloy of Ni with 39-15% Ta. They found that 5%, HCl contg. 2-3 drops HNO, was best for dissolving the Ni selid soln, without attacking the Ni, Ta. With 0-2-0-5 g, alloy in a 100-ml, beaker, dissoln, began only on warming, but then continued very slowly in the cold for 2-3 days, with occasional agitation. The residue was dried with alcohol and ether and observed microscopically. Greater amounts were prepared by electrolytic dissoln, using as anode a polished red of the alloy 50 × 3 mm. dia, in a colloid bag, centred with relation to the timplate beaker, 9 cm. high × 8 cm. wide, acting as cathode. The electrolyte was 0.75% alcoholic HCl + 20 g, citric acid + 5 g, NH₂Cl, and the c.d. 0-01 amp./cm. (at greater c.d. there was anodic oxidation, so that the product contained oxide and salta), 0-5 g, powder was obtained in 1 hr. On analysis by dissolving in HF + HNO₃, removing HF by heating with H₂SO₄, precipitating Ta with NH₂OH in the presence of NH₄Cl, and weighing as Ta₂O₅, the powder was found to contain 49-44% Ta (cf. 50-70% for Ni₃Ta theoretically).—G, V, E, T.

PYLAYEVA, Ye. II.

"An Investigation of the Structural Diagram of the Trinary System Mi - Ni 3 Nb - Ni Ta." Cand Tech Sci, Inst of Metallurgy imeni A. A. Baykov, 6 Dec 54. (VM, 24 Nov 54)

Survey of Scientific and Technical Dissertations Defended at USSR Higher Educational Institutions (11)

SO: Sum. No. 521, 2 Jun 55

USSR/ Chemistry - Metallurgy

Card

1/1

Authors

Kornilov, I. I. and Pylaeva, E. N.

Title

Study of the structural diagram of a system formed by metallic Ni3Nb-Ni₃Ta compounds.

Periodical

: . Dokl. AN SSSR, 97, Ed. 3, 455 - 457, July 21, 1954

Abstract

The study and formation of a structural diagram, for a binary system formed by metallic Ni3Nb and Ni3Ta compounds, are discussed. The study of this system was carried out by methods of thermal analysis, microstructure, specific electrohardness resistance and specific weight. Many fusions of this binary system were also subjected to x-ray analysis and the total results are described. Eight references: 6-USSR and 2-USA.

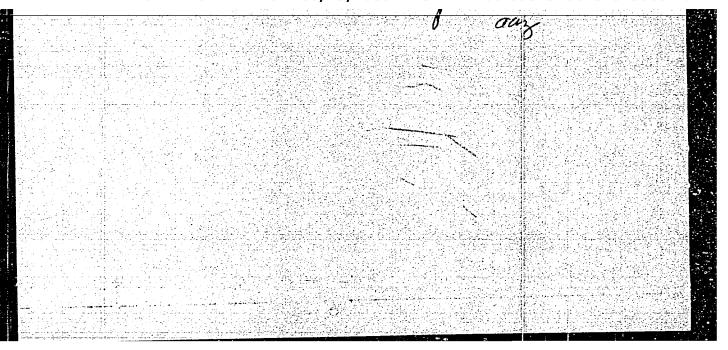
Graph, illustrations.

Institution : Acad. of Sc. USSR, The A. A. Baykov Institute of Metallurgy

Presented by: Academician I. P. Bardin, March 26, 1954

CIA-RDP86-00513R001343730004-6" APPROVED FOR RELEASE: 06/15/2000

"APPROVED FOR RELEASE: 06/15/2000 CIA-RDP86-00513R001343730004-6



PYLAYEVA VE.N.

USSR/Physical Chemistry - Thermodynamics. Thermochemistry. Equilibrium. Physico-

chemical Analysis. Phase Transitions, B-8

Abst Journal: Referat Zhur - Khimiya, No 1, 1957, 365

Author: Kornilov, I. I., and Pylayeva, Ye. N.

Institution: None

Title: Investigation of the Phase Diagram of the Ternary System Ni-Ni3Nb-

Ni₃Ta

Original

Periodical: Zh. neorgan. khimii, 1956, Vol 1, No 2, 308-316

The phase diagram for the ternary system Ni(I)-Ni3Nb(II)-Ni3Ta(III) Abstract:

was studied. A phase diagram has been constructed for the binary system formed by the metallic compounds II and III, and it is shown that it represents a continuous series of solid solutions. The phase diagram for I-II-III has been investigated along 3 radial sections from the nickel corner to the quasi-binary cross section II-III. On the

basis of the data obtained by thermic analysis, microstructure studies, and hardness and conductivity studies on the melts, it has

Card 1/2

USSR/Physical Chemistry - Thermodynamics. Thermochemistry. Equilibrium. Physicochemical Analysis. Phase Transitions, B-8

Abst Journal: Referat Zhur - Khimiya, No 1, 1957, 365

Abstract: been established that the phase diagram for I-II-III is characterized by limited solid state solubility with a continuous transition from the binary eutectic I-II to the binary eutectic of the system I-III. It is shown that the joint solubility of Nb and Ta decreases continuously as the temperature is reduced from the crystallization temperatures of the ternary alloys to room temperature.

Card 2/2

PYLAYEVA, Ye. N.

Category: USSR / Physical Chemistry

Thermodynamics. Thermochemistry. Equilibrium. Physico-

chemical analysis. Phase transitions.

B-8

Abs Jour: Referat Zhur-Khimiya, No 9, 1957, 29931

install the production of the process of the second sections where the

Author : Kornilov I. I., Pylayeva Ye. N., Volkova M. A.

Inst : Academy of Sciences USSR

15 kg/mm^).

Title : Diagram of State of Binary System Titanium - Aluminum

Orig Pub: Izv. AN SSSR, Otd. khim, n., 1956, No 7, 771-778

Abstract: Investigation of the diagram of state of Ti - Al system, by thermal, microstructure and x-ray diffraction methods, and also by means of analysis of hardness and heat-resistance. Occurence of peritectic transformations has been ascertained at 1520° (beta) + melt gamma and at 1400° (gamma + melt \supseteq Ti Al₃) and also that of a peritectoidal reaction at 1300° (beta + gamma \supseteq alpha). Solubility of Al in Ti at 1200° and 800° is, respectively, of 26 and 21.6%. Solid solutions of Al in Ti, located near the boundary of maximum solubility of Al in Ti, have highest durability at high temperature (at 550° and

Card : 1/1

-47-

PYLAYEVA, Ye.N.

137-58-4-8440

Translation from: Referativnyy zhurnal, Metallurgiya, 1958, Nr 4, p 303 (USSR)

Kornilov, I.M., Pylayeva, Ye.N., Volkova, M.A. AUTHORS:

Phase and Heat Resistance Diagram of Alloys of the Ti-Al Binary TITLE:

System (Diagramma sostava - zharoprochnost' splavov dvoynoy

sistemy Ti-Al)

Tr. In-ta metallurgii AN SSSR, 1957, Nr 2, pp 164-166 PERIODICAL:

The heat resistance and change in lattice spacing of Ti in Ti-ABSTRACT: Al alloys having up to 27.5% Al is studied. The curves of the relationship between Ti lattice spacings and Al content differ in the single-phase and double-phase regions, and the values of the a and c spacings diminish as Al content rises. The centrifugal method was employed to investigate the heat resistance, tests being run at 550°C and stresses of O = 15 kg/mm2 for 250 hours and then at 6000 and the same of for 50 hours. The specimens were made by sintering Ti powders. The criterion of heat resistance employed was the time required to attain a given bending deflection, namely, 2 and 4 mm (the latter in the case of pure Ti). The bending deflection of alloys from the region of solid Al solutions under analysis and of alloys in the heterogen-

Card 1/2

137-58-4-8440

Phase and Heat Resistance Diagram of Alloys of the Ti-Al Binary System

eous region (((3+))) rises rapidly in the process of deformation. As the concentration of Al in the solid solution rises, the bending deflection diminishes sharply (alloys with 2.5-5% Al bend 6 mm after 250 hours, while those with 7.5-20% Al bend 2-3 mm). Alloys in the biphasic region are brittle and less heat resistant than Ti and alloys from the region of solid solutions. Comparison of the curves of bending deflection for various alloys with the phase diagram and with the change in the lattice spacing shows that in the Ti-Al binary system a definite relationship exists at 550-6000 between the heat resistance, the composition, and the structure of the alloys: heat resistance exists within the bounds of a limited solid-solution range of Al content. Maximum heat resistance is observed in high-content solid Ti solutions. The compositions of alloys in the transition zone from solid solutions to the biphasic region show higher heat resistance than pure Ti, the solid solutions studied, or alloys unmistakably in the biphasic region.

V.G.

1. Aluminum-titanium alloys--Phase studies 2. Aluminum-titanium alloys--Temperature factors

Card 2/2

FYING Study of the Ni Ti - Hi Ta, Ni Ti - Ni Nb binary systems. Zhur.

Study of the Ni Ti - Hi Ta, Ni Ti - Ni Nb binary systems. Zhur.

neorg.khim. no.3:673-677 '58. (MIRA 11:4)

1.Institut metallurgii im. A.A. Baykova Akademii nauk SSSR.
(Nickel-titanium-tantalum alloys)
(Nickel-titanium-niobium alloys)

78-3-6-17/30 Kornilov, I. I., Pylayeva, Ye. N., AUTHORS: Volkova, M. A.

II. Investigations of Equilibrium in the Ternary System TITLE:

Ti-Al-Fe (II. Issledovaniye ravnovesiya v troynoy

sisteme Ti-Al-Ye)

ł

Zhurnal Neorganicheskoy Khimii, 1958, Vol. 3, Nr 6, FERIODICAL:

pp. 1391-1397 (USSR)

The ternary system Ti-Al-Fe, especially in the angle of titanium of up to 30 % of the sum Al+Fe, was investigated ABSTRACT:

by means of thermal, micro-structural - and X-ray analysis.

The alloys produced were investigated with respect to their hardness and temperature-stability. The solid solution of aluminum and iron covers a vast range in

 β -titanium at 1100°C.

The phase-compositions were investigated at temperatures of

1100, 1000, 800 and 550°C. A large part of the alloys

undergoes entectoid transition into solid solutions like in

the systems Ti-Fe: $\beta \rightarrow \alpha + \text{TiFe}$.

The occurence of the β -phase in the biphase-range α +TiFe Card 1/2

II. Investigations of Equilibrium in the Ternary System Ti-Al-Fe

78-3-6-17/30

increases according to the increase in temperatures of from 680°C to 850°C, according to the increase of the aluminumcontent in the alloys.

In the ternary system Ti-Al-Fe the p-phase dissipates at 1100°C of from 40 % to 47 % Al. The maximum solubility of iron in this phase amounts to approximately 1,5 %. A decrease in the hardness of the alloys takes place in the range of the p-solid solution in the ternary system Ti-Al-Fe. The alloys with p-phase retain their hardness when heated up to a temperature of 700°C, whereas at temperatures of from 70°C to 950°C the hardness of the alloys decreases to a smaller extent than in titanium alloys on the basis of the α-phase.

There are 17 figures, and 13 references, 4 of which are Soviet.

SUBMITTED:

June 26, 1957

AVAILABLE:

Library of Congress

Card 2/2

1. Aluminum-iron-titanium alloys--Phase studies 2. Aluminum-

iron-titanium alloys--Production

AUTHORS:

Pylaysva, Ye.N., Gladyshovakiy, Ye.I.,

SOV/ 78-3-7-28/44

Krapyakevich, P.I.

TITLE:

The Crystalline Structure of the Compounds NigNo and NigTa (Kristallinheskaya struktura soyedineniy NigNo i NigTa)

PERIODICAL:

Zhurnal neorganicheskey khimii, 1958, Vol 3, Nr 7, pp 1626-1631

(USSR)

ABSTRACT:

The metallic compounds Ni-No and Ni-Ta and 9 terrary

alloys of the series Ni₂Nb-Ni₂Ta were investigated with respect to their structure by the X-ray method. The results obtained showed that the compounds Ni₂Nb and Ni₂Ta belong to the structural type β -Cu₂Ti. The structural arrangement of atoms is the follow-

ings 2 Nb (cr Ta) in (a) with $Z_A = 2/3$

2 Ni in (b) with $Z_b = 1/3$, 4 Ni in (f) with x = 1/4; $Z_f = 1/6$. The lattice constant for the compound Ni_zNb are the following:

4 11 5.10, b = 4.24, 0 = 4.53 X

The ratio as bs on 2: 1,66: 1,78

Card 1/2

For the compound NigTa the lattice constants are as follows:

The Crystalline Structure of the Compounds NigNb and NigTa

SOV/ 78-3-7-28/44

a = 5.09, b = 4.23, c = 4.51 \times , a : b : c = 2 : 1.66 : 1.77. The emporada Ni₃Nb and Ni₃Ta together form continuous series of solid solutions. There are 2 figures, 2 tables and 5 references, 3 of which are Soulet.

ASSOCIATION:

Institut metalinegii im. A.A.Baykesa Akademii nauk SSSR 1 L'ecoshiy gosuniversitetim. I.Franko (Institute of Metallungy imen: A.A.Baykov, AS USSR and L'vor State University imen! I.Franko)

SUBMITTED:

June 18, 1957

1. Intermetallic compounds—Crystal structure 2. Intermetallic compounds—Atomic structure 3. Intermetallic compounds—X-ray analysis 4. Intermetallic compounds—Lattices

Card 2/2

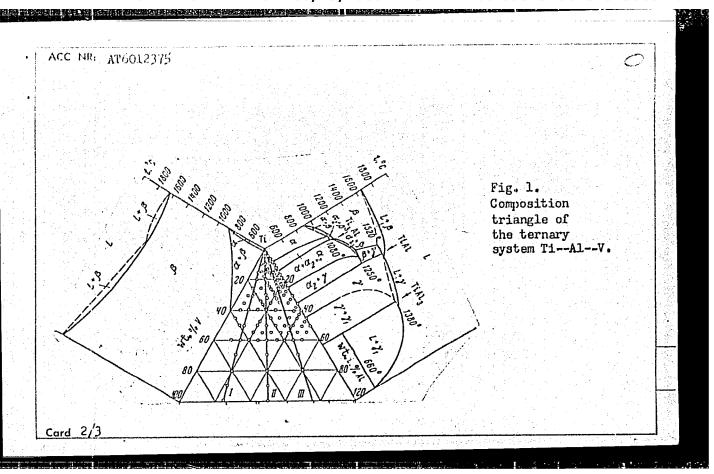
GE CHEHI MIN [No Chih-ming]; NORNILOV, I.1.; PYLAYEVA, Ye.N.

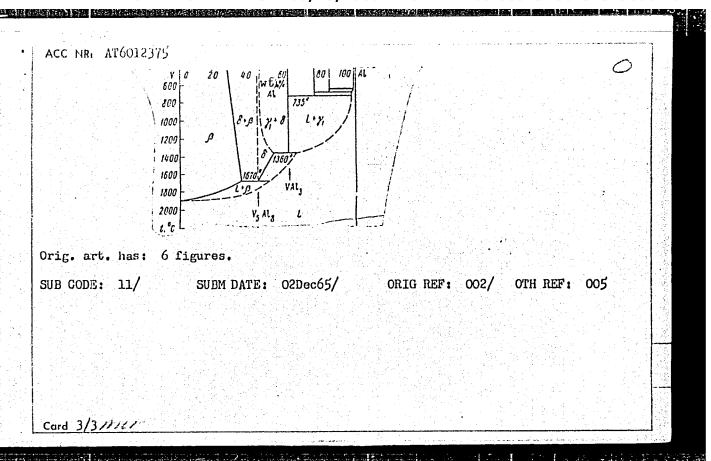
Phase equilibrium diagram of the system Ti - Al - (Mo!V=1:1).

Zhur. neorg. khim. 9 no.7:1662-1668 Jl '64. (NIRA 17:9)

SOURCE CODE: UR/0000/65/000/000/0092/0097 ACC NR. AT6012375 AUTHORS: Kornilov, I. I. (Doctor of chemical sciences, Professor); Volkova, M. A.; Pylayeva, Ye. N. for the same ORG: none TITLE: Investigation of the alloys of the ternary system Ti-Al-V SOUNCE: Soveshchaniye po metallokhimii, metallovodeniyu i primeneniyu titana i yego splavov, 6th. Hovyye issledovaniya titanovykh splavov (Hew research on titanium alloys: trudy soveshchaniva. Moscow, Izd-vo Nauka, 1965, 92-97 TOPIC TAGS: titanium, aluminum, vanadium, alloy phase diagram, ternary alloy, hardness ABSTRACT: The alloys of the system Ti-Al-V were studied. The experimental results supplement an earlier investigation by I. I. Kornilov, Ye. N. Pylayeva, M. A. Volkova, P. I. Kripyakevich, and V. Ya. Markiv (Nastoyashchiy sbornik, str. 18). The experiments were carried out with titanium iodide (99.7% Ti), AVOOO aluminum (99.99%) and carbothermal vanadium (99.5% V). The phase diagrams of the system and the microstructure, hardness, and electrical resistance of the alloys were determined. Experimental results are presented graphically (see Fig. 1). The minimum hardness and electrical resistance of alloys containing 15--16% Al and an Al/V ratio of 3:1 are due to the formation of a solid solution on the basis of the compound TiaAl in the ternary system.

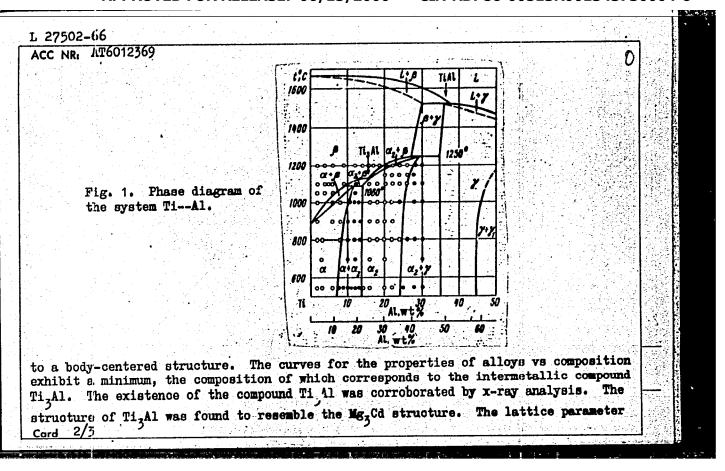
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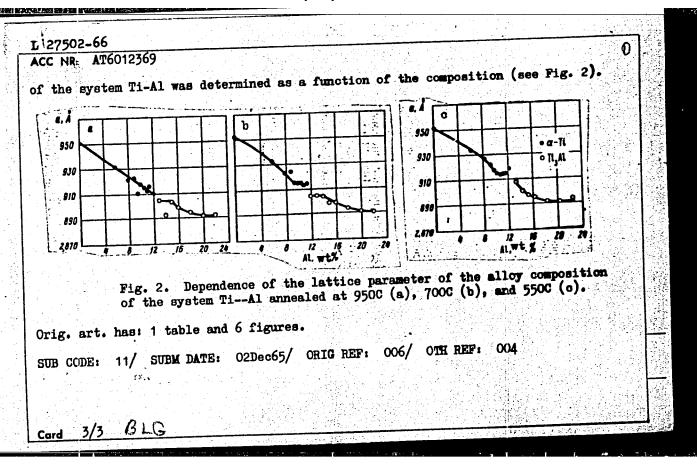




%; (m)/ENT(w)/T/ENF(t)/ETT SOURCE CODE: UR/0370/66/000/002/0137/0143 ACC NR: AP6013367 Kornilov, I. I. (Moscow); Pylayeva, Ye. N. (Moscow); Volkova, M. A. ORG: none TITLE: Evaluation of the creep of alloys of the Ti-Al-V system by the bending method at elevated temperatures SOURCE: AN SSSR. Izvestiya. Metally, no. 2, 1966, 137-143 TOPIC TAGS: creep, titanium alloy, vanadium alloy, aluminum alloy ABSTRACT: Continuing their study of the high-temperature strength of titanium alloys, the authors investigated it in the ternary system Ti-Al-V as a function of alloy composition and structure. An isothermal section of the system at 550°C was plotted on the basis of microstructural and x-ray analyses and a determination of the properties; the regions of the \prec and β solid solutions of titanium and of the intermetallic compound Ti3Al (42 phase) are indicated (see Fig. 1). Alloys of the Ti-Al-V system along sections with constant aluminum contents of 5, 7.5, and 20% were found to have a maximum creep resistance near the boundary of the limiting solutions based on α Ti, β Ti, and Ti3Al; the lowest high-temperature strength is displayed by alloys from the regions ($\alpha + \beta$) and ($\alpha + \beta$) with a coarse two-phase structure. Alloys from the region of the Y phase have a high creep resistance at Card 1/2

L 27502-66 EWT(n)/T/EWP(t)/ETI IJP(c) JH/JD/GS ACC NRI AT6012369 SOURCE CODE: UR/0000/65/000/000/0048/0055 AUTHORS: Kornilov, I. I. (Doctor of chemical sciences, Professor); Volkova, M. A.; Pylayeva, Ye. N.; Kripyakevich, P. I.; Markiv, V. Ya. ORG: none TITLE: Investigation of equilibrium diagrams of titanium-rich alloys of the system SOURCE: Soveshchaniye po metallokhimii, metallovedeniyu i primeneniyu titana i yego splavov, 6th. Novyye issledovaniya titanovykh splavov (New research on titanium alloys); trudy soveshchaniya. Moscow, Izd-vo Nauka, 1965, 48-55 TOPIC TAGS: titanium, aluminum, alloy phase diagram, titanium alloy, binary alloy, ABSTRACT: The phase diagram of the binary system Ti-Al (containing up to 30% Al) was determined. The diagram was constructed on the basis of thermal, microstructural, dilatometrical, and x-ray analysis. In addition, the specific electrical resistance and hardness of the alloy specimens were determined. The investigation supplements earlier work of N. V. Grum-Grzhimaylo, I. I. Kornilov, Ye. N. Pylayeva, and M. A. Volkova, (Dokl. AN SSSR, 1961, 137, No. 3, 599). The experimental results are summarized in graphs and tables (see Fig. 1) and compared to earlier literature data. A rearrangement takes place in the alloys in the temperature region from 882 to 1250C. These temperatures correspond to a transition from a hexagonal close-packed structure Card 1/3

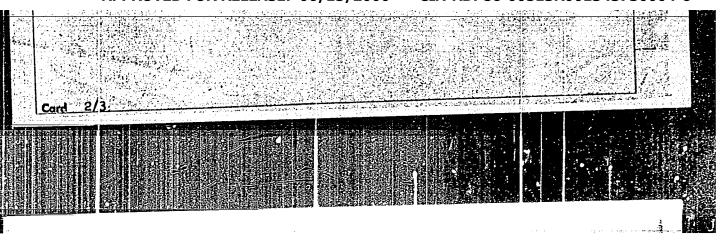




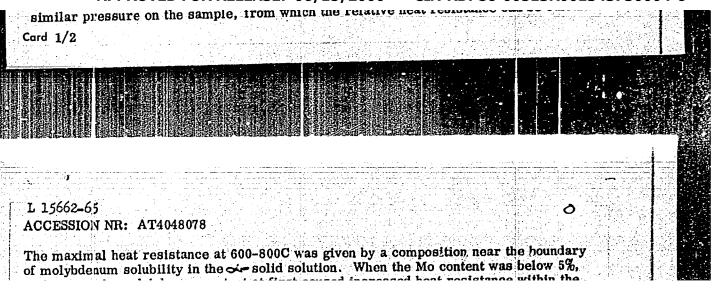
SOURCE: AN SSSR. Doklady, v. 161, no. 4, 1965, 843-846

TOPIC TAGS: titanium aluminum system, titanium alloy, aluminum containing alloy, alloy phase composition, alloy resistivity, alloy hardness

ABSTRACT: Binary Ti-Al alloys containing from 0 to 30% Al, levitation melted correct melted in an inert gas atmosphere, were investigated in as-cast condition or deformed at 800—1000C with a reduction of 30%. The thermal analysis data showed that all alloys undergo the solid state transformation from a c.p.h. to b.c.c. structure. Microscopic examination and x-ray diffraction patterns revealed the following phases, (solid solutions): β—on a β-Ti base, α—on a α-Ti base, α—on a base of the ordered tetragonal structure of Ti Al compound of the Magach type. Results of the measurements of the resistivity and hardness closely corresponded to one another and confirmed the results of the thermal, metallographic, and x-ray analysis. A phase diagram of the investigated Ti-Al system based on the results obtained is shown in Fig. 1 of the Enclosure. Orig. art. has: 3 figures.



ASD(m)-3 JD/JG/MLK 8/0000/64/000/000/0236/0239 ACCESSION NR: AT4048078 37+1 AUTHOR: Py*layeva, Ye. N., (Candidate of technical sciences); Ko, Chih-ming TITLE: Heat resistance of Ti-Al-Mo-V alloys SOURCE: Soveshchaniye po metallurgii, metallovedeniyu i primeneniyu titana i yego splayov. 5th, Moscow, 1963. Metallovedenlye titana (Metallography of fitanium); tiudy soveshchaniya. Moscow, Izd-vo Nauka, 1964, 236-239 TOPIC TAGS: titanium alloy, titanium alloy heat resistance, aluminum containing alloy, molybdenum containing alloy, vanadium containing alloy ABSTRACT: The authors investigated the heat resistance of Ti-Al-Mo and Ti-Al-(Mo: V=1:1) alloys depending on composition and structure. The hot hardness method and centrifugal bending method were used. For the first test, a load of 1 kg was used with 1, 5, 10 and 20 minutes of deformation. The bending stress for the second method was 550 to 800C depending on the aluminum



ASSOCIATION: none
SUBMITTED: 15Jul64 ENCL: 00 SUB CODE: MM
NO REF SOV: 005 OTHER: 000

Cord 2/2

L 14307-65 EPF(n)-2/EPP/ MT(m)/EWP(b)/EWP(t) Ps-4/Pu-4 ASD(m)-3/AFTC(p) WW/JD/JG/MLK

ACCESSION NR: AT4048049 S/0000, 64/000/000/0038/0042

AUTHOR: Py*layeva, Ye. N., Volkova, M. A.

TITLE: A study of the alloys of the ternary Ti-Al-Zr system

SOURCE: Soveshchaniye po metallurgii, metallovedeniyu i primeneniyu titana i yego splavov. 5th, Moscow, 1963. Metallovedeniye titana (Metallography of itanium); trudy soveshchaniya. Moscow, Izd-vo Nauka, 1964, 38-42

TOPIC TAGS: alloy structure, alloy phase composition, titanium alloy, aluminum alloy, zirconium alloy, alloy hardness

ABSTRACT: Although the Ti-Al-Zr system should produce a broad cross section of solid solutions which could be the bases for high-temperature alloys, data on this subject are totally lacking. For the preparation of samples, sponge titanium, aluminum and zirconium of the highest purity were used. The Ti-Al ratio was kept at 6:1 to facilitate the formation of Ti₆Al. The amount of Ti₆Al was varied from 100% by weight to 0, while the amount of zirconium was increased from 0 to 100%. Samples were heated to temperatures ranging from 1200 to 500C and held there for periods ranging from 6 to 750 hours, respectively. Heating was done by an arc furnace in an argon atmosphere. Since the

Card 1/2

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CIA-RDP86-00513R001343730004-6

L 14307-65

ACCESSION NR: AT4048049

0

weight difference of the samples was never more than 0.5%, no chemical analysis was performed. Microstructural and thermal analyses were performed on each sample. The specific electrical resistance and hardness were determined. The results of tests on samples containing the usual 6:1 ratio of titanium to aluminum and having less than 10% by wt. zirconium, when heated to 500C, showed that the samples contained a new solid solution of $\alpha + \beta_2$ form. All alloys of this type undergo polymorphic transformations, analogous to the transformation of pure titanium and zirconium in the Ti-Zr system, in which the transformation may be pinpointed at the minimum on the temperature vs. composition curve, i.e. at 660C and 65% Zr. The specific electrical resistance and hardness, determined from tests on samples which were quenched from temperatures

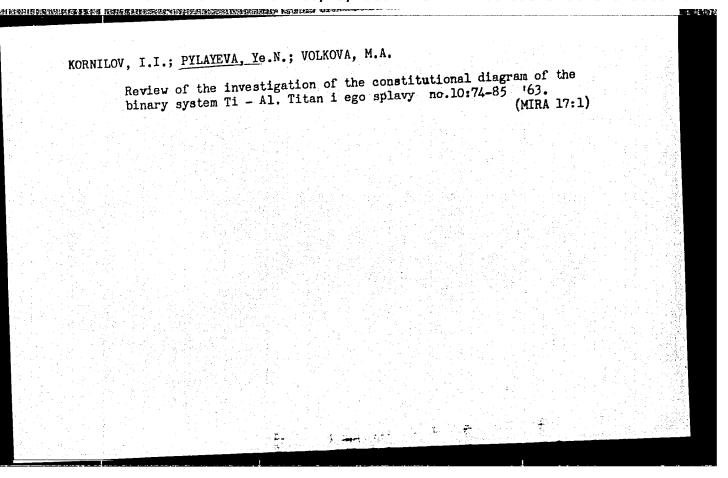
in samples cooled from a composition which produces a p-solid solution. Olig. also has: 4 graphs, 1 table and 5 photomicrographs.

ASSOCIATION: None

SUBMITTED: 15Jul64 ENCL: 00 SUB CODE: MM

NO REF SOV: 002 OTHER: 004

Card 13/2



GE CHZHI-IN [Ko Chih-ming]; PYLAYEVA, Ye.N.

Investigating the phase equilibrium in the system Ti - Al - Mo in the region of titanium-rich alloys. Titan i ego splavy no.10:14-21

Investigating phase transformations in the system Ti - Al - Mo. [bid.:22-26] (MIRA 17:1)

KORNILOV, I.I. (Moskva); SHINYAYEV, A.Ya. (Moskva); PYLAYEVA, Ye.N. (Moskva)

Greep of certain metal compounds. Izv. AN SSSR. Met. 1 gor. delo no.5:113-115 S-0 '63. (MIRA 16:11)

\$/078/63/008/002/003/012 B101/B186

AUTHORS:

Ko Chih-ming, Kornilov, I. I., Pylayeva, Ye. N.

TITLE:

Investigation of the phase diagram of the system titaniumaluminium-molybdenum in the titanium-rich alloying regions

Zhurnal neorganicheskoy khimii, v. 8, no. 2, 1963, 366 - 372 PERIODICAL:

TEXT: The present study belongs to a series of investigations of the quaternary system Ti-Al-Mo-V. In order to obtain missing data the solidus isotherms of alloys of the system Ti-Al-Mo containing (% by weight) 55 - 95 Ti, 5 - 35 Al and 0.5 - 40 Mo were plotted. Using these and data relating to the microstructure and X-ray analysis, nine polythermal crosssections and three isothermal cross-sections were plotted. Results: Alloys rich in titanium melt at 1700°C. The m.p. wises to 2000°C with 50% Mo, whereas it falls to 1400°C with high aluminium content. Increasing molybdenum content causes the temperature of the a = \$ transitions to drop, increasing aluminium content raises it. With 5 to 10% Al content the polythernal cross-section passes through the crystallization regions of the β -, $(\alpha+\beta)$ - and α -phases. With 15 to 20% Al content, the β -phase is the

Card 1/3

S/078/63/008/002/003/012 B101/B186

Investigation of the phase ...

first to crystallize, which is then partially converted into the α -phase and finally the w-phase separates itself from the a-phase. With 25% Al the $(\alpha+\beta+\gamma)$ region widens and a $(\beta+\gamma)$ region occurs. With 30% Al, the γ -phase is separated from the β -phase, which crystallizes first, and then the α -phase is formed due to peritectic transition. With 35% Al the β-phase crystallizes first and is followed by the χ -phase, so that a $(\beta+\chi)$ region is formed. With 40% Al only the x-phase forms from the melt. In the isothermal crosssection at 1100°C the largest region is the one of the β-phase reaching up to 10% Al. The α-phase, forming a narrow strip is adjacent to the Ti-Al side between 10 and 25% Al. The TiAl-based ternary solid solution, the r-phase, has only a small region. The maximum solubility of Mo in TiAl is about 11 - 12% at 1100°C. Titanium alloys with less than 12% Al + Mo show martensitic structure after quenching in water. At 800°C the β-region becomes smaller and its boundary is displaced towards the Ti - Mo side. The central part of the cross-section is formed by the (a+\$)-phase. At 6000C the $(\alpha+\beta)$ region and the $(\alpha+\beta+\chi)$ region widen. At this temperature the solubility of molybdenum in the solid a solution is about 1.0%. Between 600 and 1100°C the following phases are in equilibrium with one another: $\alpha, \beta, \gamma, \alpha+\beta, \alpha+\gamma, \beta+\gamma, \gamma+TiAl_{\chi}, \alpha+\beta+\zeta$ and others. There are 6 figures and 1 table. The English-language references are: H. D. Kessler, Armour Card 2/3

Investigation of the phase ...

s/078/63/008/002/003/012 B101/B186

Research Foundation, Report on Contract No AD 11-022 ORD to Watertown Arsenal, 1951; H. Margolin et al., New York Univ. Eng. Res. Div., Final Report in Watertown Arsenal Laboratory, 1954, on Contract No Da-030-069-

ASSOCIATION: Institut metallurgii im. A. A. Baykova Akademii nauk SSSR (Institute of Metallurgy imeni A. A. Baykov of the Academy

SUBMITTED:

June 6, 1962

Card 3/3

GE CHZHI-MIN [Ko Chih-ming] (Moskva); KORNILOV, I.I. (Moskva); PYLAYEVA, Ye.N. (Moskva)

Investigating the structure and properties of alloys in the system titanium - aluminum - molybdenum. Izv.AN SSR. Otd.tekh.nauk.

Met.i topl. no.4:114-118 J1-Ag '62. (MIRA 15:8)

(Titanium-aluminum-molybdenum alloys-Thermal properties)

5/180/62/000/004/004/009 Kornilov, I.I., Pylayeva, Ye.N. (Moscow) Investigation of the structure and properties of 18.8200 19 8100, Ko-Chih-Ming, titanium-aluminium-molybdenum alloys PERIODICAL: Akademiya nauk SSSR. Izvestiya. Otdeleniye AUTHORS: Akademiya nauk Sook. Izvestiya. Utdeteniye tekhnicheskikh nauk. Metallurgiya i toplivo, TITLE: Using the hot-hardness technique for a rapid assessment of the alloy properties as a function of temperature, and of the hot-hardrage and organization and of the hot-hardrage and organization. examination was made of the hot-hardness and creep of titaniumcorner alloys of the Ti-Al-Mo ternary system along sections corner alloys of the il-Al-Pio ternary system along sections parallel to the Ti-Mo side of the concentration triangle at parallel to the li-mo side of the concentration triangle at aluminium contents of 0, 5, 10, 15, 20 and 36% and at molybdenum aluminium contents of 0, 5, 10, tests were made in BMM-1M (VIM-1M) contents from 0 to 10%. The tests were made in BWM-1M (VIM-1M) The test specimens were melted in an arc-furnace with a non-consumable tungsten electrode in an argon atmosphere with a non-consumable tungsten electrode in an argon atmosphere and were vacuum-annealed at 1100°C for 24 hours, then annealed at 100°C and finally cooled with the furnace contents from 0 to 10%. and were vacuum-annealed at 1100 C for 24 nours, then annealed again for 24 hours at 600°C and finally cooled with the furnace. The hardness (1 kg load) was determined in the interval 20 to The narquess (1 kg road) was determined in time of 1 minute. Card 1/3

Investigation of the structure ...

S/180/62/000/004/004/009 E040/E435

hardness of titanium and of its alloy with 5% Al dropped progressively with rising temperature, whereas the hardness of alloys with 10, 15 and 20% Al changes little up to about 700 to Molybdenum additions have a much less beneficial effect on the hardness of titanium, especially at high temperatures: the hardness of binary titanium alloys with up to 5% Mo decreased with rising temperature. The hardness of titanium remained unchanged as the temperature increased to 500 to 600°C if the molybdenum content was raised to 10%. Studies of the effect of molybdenum additions on the hardness of Ti-Al alloys showed that the hardness at room temperature rises when the Mo content is from 3 to 10%; at higher temperatures the hardness drops. The creep of the alloys was examined at 700°C using a method described previously (Osipov, K.A., T'ien-te-Cheng. Izv.AN SSSR.OTN. M i T., Molybdenum concentrations up to 1 - 3% increase the no.4, 1959). resistance of titanium to plastic deformation at 700°C but this effect disappears almost completely if the molybdenum concentration is raised to 10%. In ternary Ti alloys (with 5, 15 and 20% A1), the highest heat resistance at 700°C was observed in alloys with Card 2/3

Card 3/3

DEOR RELEASE: 06/15/2000 C

CIA-RDP86-00513R00134373000

KRIPYAKEVICH, P.I.; GLADYSHEVSKIY, Ye.I.; PYLAYEVA, Ye.N.

Compounds of the type W6Fe7 in the systems Ta - Ni and Nb - Ni.
Kristallografiia 7 no.2:212-216 Mr-Ap '62. (MIRA 15:4)

1. L'vovskiy gosudarstvennyy universitet imeni I.Franko.
(Tantalum-nickel-niobium alloys) (Crystallography)

KRIFYAKEVICH, P.I., FYLAYEVA, Ye.N.

Crystal structure of Ta2Ni. Zhur.strukt.khim. 3 no.1:35-37
Ja-F '62.

1. L'vovskiy gosudarstvennyy universitet imeni Iv.Franko i
Institut metallurgii imeni A.A.Baykova AN SSSR.

(Tantalum-nickel alloys) (Crystallography)

KORNILOV, I.I.; PYLAYEVA, Ye.N.

Phase diagram of the tantalum - nickel system. Zhur.neorg.khim.
7 no.31590-595 Mr 162. (MIRA 15:3)
(Nickel-tantalum alloys) (Phase rule and equilibrium)

S/598/62/000/007/008/040 D267/D307

AUTHORS: Pylayeva, Ye. N. and Volkova, M. A.

TITLE: Solubility of silicon in \alpha-titanium

SOURCE: Akademiya nauk SSSR. Institut metallurgii. Titan i yego

splavy. no. 7, Moscow, 1962. Metallokhimiya i novyye

splavy, 74-77

TEXT: Seven compositions of binary alloys (Si content range 0.1 - 2 wt-%) were investigated at 850, 800 and 600°C, using the methods of micro-structural analysis and hardness. The alloys obtained by levitation melting were subsequently heat treated. It was found that the solubility of Si in Ti amounts to 0.40 wt-% at 850°C, 0.35 wt-% at 800°C, and 0.30 wt-% at 600°C. The presence of the compound Ti_Si_z in alloys with 0.5, 0.75 and 1% Si was borne out by the phase analysis of intermetallic compounds. The increase of Si content increases the hardness and strength of the alloy, with a simultaneous reduction of plasticity. There are 4 figures and 2 tables.

Card 1/1

s/598762/000/007/011/040 D244/D307

12.1285

AUTHORS: Kornilov, I. I., Pylayeva, Ye. N. and Volkova, M. A.

TITLE: Properties of the alloys of the ternary titanium-alumi-

num-vanadium system

SOURCE: Akademiya nauk SSSR. Institut metallurgii. Titan i yego

splavy. no. 7, Moscow, 1962. Metallokhimiya i novyye

splavy, 89-94

TEXT: The work is a continuation of previous investigations of Ti-Al and Ti-Al-Fe alloys. In this investigation the heat stability of Ti rich alloys of ternary system Ti-Al-V was investigated. Microstructure of the alloys at 600° C included either one α -phase or two phases α and $(\alpha + \beta)$. The alloy with 7.5% Al and 0.5% V had a single phase structure of α -solid solution and the alloy with 7.5% Al and 4% V consisted of $(\alpha + \beta)$ phases. The heat stability was determined by the method of centrifugal bending under a tension of 15 kg/mm² at 550°C. For alloys containing 5% Al, additions of V from 0.5 to 1% did not decrease their heat stability. Further in-

Card 1/2

Properties of the alloys ...

S/598/62/000/007/011/040 D244/D307

creases of V from 5 to 10% led to the formation of two phases α - β , which decreased the heat stability. Influence of V on the alloys with 7.5% Al was similar. A number of alloys was prepared by powder metallurgy and tested for heat stability. The most heat-stable alloys contained 10% Al and 30% V or 15% Al and 15% V. It was shown that the addition of Al (0 - 15%) to the alloys with a constant content of V (2, 3, 4, 5%) increased their heat stability. There are 7 figures.

Card 2/2

s/598762/000/007/018/040

Kornilov, I. I., Mikheyev, V. S., Pylayeva, Ye. N., Volkova, M. A., Borok, B. A., Shchegoleva, R. P. and Golubeva, L. S. 12 1275 AUTHORS:

The effect of aluminum on the structure and properties of TITLE:

a Ti-Al-Cr-Fe-Si-B alloy prepared by powder metallurgy

Akademiya nauk SSSR. Institut metallurgii. Titan i yego SOURCE:

splavy. no. 7, Moscow, 1962. Metallokhimiya i novyye

splavy, 130-134

TEXT: The authors studied the effect of varying amounts of Al in Ti-Al alloys (1 - 7% by weight Al) and in alloys of the Ti-Al-Cr-Fe-Si-B system (1.5 - 12% by weight Al) on the structure and properties of the alloys. Strength of the Ti-Al alloys increased from 77.2 to 107-3 kg/mm² as the Al content rose from 0 to 7%; the strength of alloy AT4 (AT4) increased from 104 to 142 kg/mm² as the Al content rose from 1.5 to 10%. Plasticities of the alloys de-creased and the heat resistance of AT4 increased as the aluminum

Card 1/2

The effect of aluminum ...

S/598/62/000/007/018/040 D290/D307

contents became higher. The rate of oxidation of AT4 in air at 700°C decreases by about 60% as the Al content rose from 5 to 12% by weight. There are 4 figures and 4 tables.

Card 2/2

62/000/007/019/040

18,1225

AUTHORS:

Kornilov, I. I., <u>Pylayeva, Ye. N., Volkova, M. A.,</u> Borok, B. A., Shchegoleva, R. P. and Golubeva, L. S.

The effect of silicon on the properties of a 6-component TITLE:

alloy of the system Ti-Al-Cr-Fe-Si-B prepared by powder

metallurgy

Akademiya nauk SSSR. Institut metallurgii. Titan i yego SOURCE:

splavy. no. 7, Moscow, 1962. Metallokhimiya i novyye

splavy, 136-139

The authors studied the effect of varying amounts of silicon in Ti-Si alloys and in alloys of the system Ti-Al-Cr-Fe-Si-B on the properties of the alloys, in order to find the optimum Si concentration in alloy A74 (AT4). The mechanical properties were measured in both the forged and hot worked conditions. The strength of the Ti-Si alloy increased from 77.2 to 100.8 kg/mm² as the Si content increased from 0 - 2% while the strength of the alloy AT4 increased from 110 to 138 kg/mm2 with the addition of 1.5% Si. Pla-

Card 1/2

s/078/62/007/003/008/019 al₁862 B110/B138 Kornilov, I. I., Pylayeva, Ye. N. Constitution diagram of the tantalum - nickel system Zhurnal neorganicheskoy khimii, v. 7, no. 3, 1962, 590-595 AUTHORS: TEXT: From the results of this study the complete constitution diagram of That: rroll the results of this study the complete constitution diagram the binary system tantalum - nickel was constructed. 10 g of tantalum the binary system tantalum - nickel was induction malted in avenage (00 gd) and 10 g of U-(N-0) nickel were induction malted in avenage. the binary system tantalum - nickel was constructed. 10 g of tantalum (99.8%) and 10 g of H-C(N-0) nickel were induction melted in suspension in a purified He atmosphere. The resulting allows were submitted to the purified He atmosphere. TITLE: purified He atmosphere. The resulting alloys were submitted to thermal, PERIODICAL: microstructural, and X-ray structural analyses, and hardness microstructural, and A-ray Structural analyses, and margness results of the Ta-rich alloys were measured on an optical solidus temperatures of the Ta-rich alloys were measured on an optical recommendation on a Kurnakov nyrometer and those of allow crystallization on a Kurnakov nyrometer. Tyrometer, and those of alloy crystallization on a Kurnakov pyrometer fyromever, and those of alloy crystallization on a kurnakov pyromet and with non contact thermography. The liquidus has six branches: (3) Talli; and with non contact thermography. The liquidus has six oranges.

(2) Ta 2 Ni;

crystallization of the \$ solid solution on a Ni base. The The liquidus (4) TaNi2; (5) TaNi3, and (6) a solid solution on a Ni base. branches intersecting at 1785, 1570, and 1420°C correspond to the following neritectic equilibria: () + melt = me Ni. me Ni melt = me Ni. melt + TaNiz peritectic equilibria: (3 + melt = Ta2Ni; Card 1/3

S/078/62/007/003/008/019 B110/B138

Constitution diagram of the ...

≥ TaNi; and at 1320 and 1360°C, to the eutectic reactions: melt = TaNi + TaNi2; melt = TaNi3 + Ni solid solution. Three new compounds were detected: Ta, Ni (66.6 atomic % or 86.25 weight % Ta); TaNi (50.0 atomic % or 76.05 weight % Ta); TaNi2 (33.3 atomic % or 60.88 weight % Ta). The microstructure was examined in cast alloys quenched from 1600, 1500, 1400, 1300, and 1200°C (soaking time 100 hrs each), and others annealed for 50 hrs at 1110°C, 100 hrs at 1000°C, and 250 hrs at 800°C. At 94 atomic % Ta after quenching from 1600°C, polyhedra of a solid solution were formed. At 95.89 atomic % Ta and quenching from 15000 a second phase was precipitated within and on the grain boundaries. At 99.62 atomic % Ta and quenching from 1300°C the solid solution began to disintegrate. Ta, Ni is formed from the peritectic reaction which takes place at 80.10% Taafter quenching from 1300°C. Ta Ni (66.6 atomic % Ta) has a dendritic structure in the cast state which changes into polyhedral after prolonged annealing at a high temperature. At 60.44% Ta, there is a peritectic reaction between melt and Ta, Ni with formation of Talli. After prolonged annealing Talli (50 % Ta) assumes a polyhedral structure. A eutectic reaction occurs between TaNi and TaNi 2 in the Card 2/3

S/078/62/007/003/008/0 B110/B138

Constitution diagram of the ...

alloy with 43.7 % Ta quenched from 1200°C. The alloy with 35.12 % Ta displayed light TaNi grains on the eutectic background. TaNi, with 33.3% Ta has polyhedral structure after prolonged annealing. The microstructural and thermal analyses thus show that the compounds of the system neither dissolve, nor form solid solutions with their components. X-ray diffraction patterns taken with $\operatorname{Cr} K_{\infty}$ radiation fitted the microstructural data. Vickers hardness tests were made on cast, annealed and quenched (1200°C) alloys at 10 kg/mm². The addition of Ni to Ta increases hardness from 135 (pure Ta) to 847 (cast and annealed). The maxima of 847 and 627 are for Ta2Ni and TaNi. The minima at 342 and 322, for TaNi and TaNi, but this is still higher than the values for their components: 135 for Ta and 60 for Ni. P. I. Kripyakevich is thanked for his X-ray structural analyses. There are 6 figures, 1 table, and 6 references: 4 Soviet and 2 non-Soviet: The two references to English-language publications read as follows: E. Therkelsen. Metals Alloys, 4 105 (1938). M. Hansen, Constitution of binary alloys, Mc Graw-Hill Book Company. New York, Toronto, London, 1958, p. 1045.

SUBMITTED: March 6, 1961

Card 3/3

5 2610 4016

S/192/62/003/001/001/002 D258/D303

AUTHORS:

Kripyakevich, P.I. and Pylayeva, Ye.N.

TITLE:

The crystal structure of Ta Ni

PERIODICAL:

Zhurnal strukturnoy khimii, v. 3, no. 1, 1962, 35-37

TEXT: The authors confirmed by x-ray analysis the existence of $TaNi_{2}$? TaNi (or a compound with a composition near to it), and $Ta_{2}Ni$; they also defined the crystal structure of the latter. The 3 compounds have been identified by I.I. Kornilov and Ye.N.Pylayeva (Ref. 5: Zh.neorg.khimii, in press), being formed in the following reactions: (1) $TaNi_{2} \rightleftharpoons \ell + Ta_{1} = (1420^{\circ}C)$; (2) $TaNi \rightleftharpoons \ell + Ta_{2} = (1570^{\circ}C)$; and (3) $Ta_{2}Ni \rightleftharpoons \ell + \beta = (1770^{\circ}C)$, where β is a solid solution of both metals. Specifically, log samples of alloys were prepared by induction melting in an atmosphere of purified He. Homogeneous structures and compositions were ensured by firstly, using 99.6% pure Ta and C.00-grade Ni, secondly by

Card 1/4

S/192/62/003/001/001/002 D258/D303

The crystal ...

avoiding the use of crucibles in melting and thirdly by carefully controlling the composition of charges. The alloys were homogenized for 1000 hrs. at 800 C, prior to their x-ray analysis. The latter proved the existence of the 3 compounds at 8000 C. X-ray powder photography (Crexistence of the 3 compounds at 8000 C. X-ray powder photography (Crexistence of the 3 compounds at 8000 C. X-ray powder photography (Crexistence of the 3 compounds at 8000 C. X-ray powder photography (Crexistence of the 3 compounds at 8000 C. X-ray powder photography (Crexistence of K. C. -radiation) of Ta Ni indicated a tetragonal body-centered lattice, with the constants a 2 6.216 $^\pm$ 0.005 Å, c = 4.872 $^\pm$ 0.004 Å; c/a = 0.784. These constants are similar to those of Ta₂Si, thus indicating for Ta₂Ni a structure of the CuAl₂ type (space group 14/mcm - D_{4h}; 4Ni in 4(a)00 lbs. The interatomic distances and coordination 0.167; it was accurately defined by photometry of lines 411, 402, 332, and 004 and found equal to 0.158. The interatomic distances and coordination numbers are given, as follows:

Card 2/4

S/192/62/003/001/001/002 D258/D303

| Atoms | d(Å) Coordeno |
|---------------|-------------------|
| Ni-2Ni 8Ta | 2.44) 2.64) 10 |
| Ta-4Ni | 2.64) |
| lTa 2Ta | 2.78) 15 2.92) |
| 4Ta | 3.31) |

There are 2 tables, 1 figure and 13 references: 4 Soviet-bloc and 9 non-Soviet-bloc. The 4 most recent references to the English-language publications read as follows: N.Karlsson, J.Inst.Metals, 79, 391 (1951); J.R.Murray, J.Inst.Metals, 84, 4, 91 (1955); P.Duwez and J.L. Taylor, J.Metals, 2, 9, 1173 (1950); and J.S. Kasper and R.M. Waterstrat. Acta crystallogr. 9, 3, 289 (1956).

ASSOCIATION: L'vovskiy gosudarstvenny universitét im. Iv. Franko (Lvov State University im. Iv. Franko); Institut metallurgii

Card 5/4

The crystal ...

33707

The crystal ...

5/192/62/003/001/001/002 D258/D303

im. A.A. Baykova AN SSSR (Institute of Metallurgy, im. A.A. Baykov, AS

SUBMITTED:

March 2, 1961

Card 4/4

KORNILOV, I.I.; BUDBERG, P.B.; VOLKOVA, M.A.; PROKHANOV, V.F.;

PYLAYEVA, Ye.N.

Developing a method of hot pressing of titanium and titanium alloy powders. Titan i ege splavy no. 1:25-32 '58. (MIRA 14:5)

1. Institut metallurgii AN SSSR. (Titanium-Metallurgy) (Powder metallurgy)

KORNILOV, I.I.; PYLAYEVA, Te.N.

Equilibrium in the ternary system of themetallides Ni Nb - Ni Ti - Ni Ta. Izv. AN SSSR. Otd. khim. nauk no.2:197-200 F 3 (61, 3)

(Nickel compounds) (Tantalum compounds) (Titanium compounds) (Niobium compounds)

<u> Marakilari Balada Jalabalira da</u>

GRUM-GRZHIMYLO, N.V.; KORNILOV, I.I.; PYLAYEVA, Ye.N.; VOLKOVA, M.A.

Metallic compounds in the region of Asolid solutions of the system titanium - aluminum. Dokl AN SSSR 137 no.3:599-602 Mr '61.

1. Institut metallurgii im.A.A.Baykova AN SSSR. (MIRA 12:2)
I.I.Chernyayevym.

(Titanium-aluminum alloys)

18.1285

also 1555

~S/020/61/137/003/018/030 B103/B208

AUTHORS:

Grum-Grzhimaylo, N. V., Kornilov, I. I., Pylayeva, Ye. N.,

and Volkova, M. A.

TITLE:

Metallic compounds in the range of solid α -solutions of

the system titanium-aluminum

PERIODICAL:

Doklady Akademii nauk SSSR, v. 137, no. 3, 1961, 599-602

TEXT: The authors proved (Ref. 6: Tr. inst. metallurgii AN SSSR, no. 2, 1957) that in titanium - aluminum alloys (7.5-20 wt% Al) the resistance to creeping in bending deformation by the centrifugal method rapidly increases as plasticity decreases. They point out that such a change of properties in the range of solid solutions of the binary system Ti - Al could not be explained by conventional methods of metallographic analysis. The objectives of the present study were therefore the following:

1) investigation of the range of solid a-solution in the Ti - Al system;

2) determination of the nature of phases appearing in it by measuring the Hall effect as a function of the composition of the alloys. The authors have previously proved (Ref. 9: ZhNKh, 2, no. 10, 1957; Ref. 10: ibid,

Card 1/8

Metallic compounds in the range of ...

S/020/61/137/003/018/030 B103/B208

31, no. 9, 1956) that the galvanomagnetic effects are related to the composition of various alloys in a way that salient points and jumps appear in the diagram composition-versus-Hall effect. This phenomenon can be explained by the fact that the electron states in the outer atomic shells are changed by applying a magnetic field. This affects the behavior of conduction electrons and alters the values of the Hall constant. The galvanomagnetic effects are closely related to the behavior of the electron components of the outer atomic shells. The state of the outer shell may be studied with high accuracy on the basis of these effects. The character of the chemical bond between various atoms of metallic alloys may thus be explained. The authors prepared alloys from pure titanium and aluminum with an Al content up to 40 wt% by two methods: 1) powder metallurgy by pressing and sintering in vacuo at 600-1000°C for 50-100 hr. 2) melting in the arc furnace with a wear-resistant tungsten electrode. The current collectors were triangular and knifeshaped at the point of contact with the specimen. They glided along the polished lateral faces of the sample by means of micrometer screws. method and measuring apparatus are described in Ref. 11 (N. V. Grum-Grzhimaylo, ZhNKh, 3, no. 7, 1958). Table 1 contains the resultant mean Card 2/8

Metallic compounds in the range of ...

S/020/61/137/003/018/030 B103/B208

values of the Hall constant of the alloys. On the basis of these data, the authors plotted a diagram of this constant as a function of the composition (Fig. 1). Two (a and b) jumps from the linear variation of the Hall constant to another linear variation are seen. These jumps correspond to: a) the compound Ti Al with 14.3 atom% (9 wt%) of aluminum; b) the compound Ti Al with 25 atom% (16 wt%) Al. The sintered and the cast alloys showed the same behavior. The cast alloys were subjected to homogenizing heat treatment (between 600 and 900°C for 200-350 hr) immediately after measuring the Hall constant. The limited range of the solid α -solution offers considerable difficulties in the presence of two metallic compounds if the order of variations of the Hall constant has to be determined. This determination requires an increased precision of measurement which was achieved by the device applied here. The authors conclude from their data that the solid aluminum solutions in α -titanium exhibit a complicated kind of interaction owing to the existence of the two compounds Ti 6Al and Ti 3Al which apparently have a hexagonal lattice. They might result from colid calutions

They might result from solid solutions and correspond to compounds of the Kurnakov type (Ref. 12: I. Kornilov, Izv. AN SSSR, OKhN, 1957,

Card 3/8

Metallic compounds in the range of ...

S/020/61/137/003/018/030 B103/B208

no. 4,.395). The diagrams of the Hall constant in the range of the γ-phase in alloys with 46.16 atom% (33wt%) to 53.85 atom% (40.0 wt%) aluminum show a sharp discontinuity at 50.0 atom% (36.02 wt%).aluminum. It corresponds to the compound TiAl which was detected by other methods of physicochemical analysis. The equilibrium of the compounds Ti₆Al, Ti₃Al,

TiAl and the proof of their existence in the phase diagram depend on the kinetics and on the conditions of their formation which have to be further studied. The appearance of these compounds in the system Ti - Al increases the heat resistance of the alloys and rapidly decreases their plasticity at an aluminum content of more than 7-8 wt%. There are 1 figure, 1 table, and 12 references: 8 Soviet-bloc and 4 non-Soviet-bloc. The reference to the English-language publication reads as follows:

M. Hansen, Constitution of binary alloys, N.Y. London, 1958, p. 139 (Ref. 1).

ASSOCIATION:

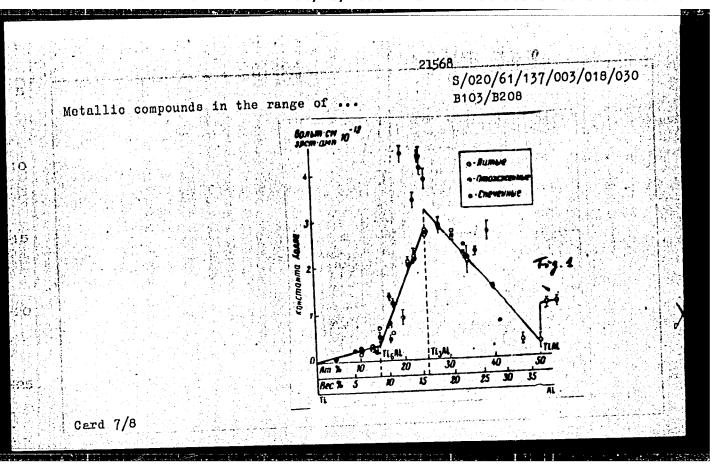
Institut metallurgii im. A. A. Baykova Akademii nauk SSSR (Institute of Metallurgy imeni A. A. Baykov of the Academy of Sciences USSR)

Card 4/8

| Metallic compounds | in the range of | 21568 S/020/61/1 B103/B208 | 37/003/018/030 | |
|---------------------|-----------------------------------------------------------------------------|----------------------------------------------------|-------------------|--------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------------|
| PRESENTED: Octo | ber 27, 1960, by I. I. | Chernyayev, Academ | ician | |
| SUBMITTED: Octo | ber 7, 1960 | | | |
| alloys, 2) sintered | 1) Al content, %, 2)-4, , 3) cast, 4) cast and | annealed. | | 1 |
| | одержание АІ, % | Константа Холла х 10-18 для сплав | BMH492ROTO ELITHR | A Comment of the Comm |
| | одержание Al, % свеченимя 4,36 а 0,023035 ± 0,026 2,5' в 0,223000 ± 0,015 | Константа Холла х 10-10 для сплав | ANTES OTOMERSHIME | The second of th |
| | одержание Al. % свечениых | Антыя Антыя 0 0,15484 ± 0,0150 0,170309 ± 0,045 | | |

"APPROVED FOR RELEASE: 06/15/2000 CIA-RDP86-00513R001343730004-6

| Metallic compo | unds in the | range | of | 21568 S/020/61/1 B103/B208 | 37/003/018/030 |) |
|----------------|--------------------------------------------------------------------------|-------|---------------------------------------------------------|-------------------------------------------------------------------|-------------------------------------------------|---|
| Metalite compo | 20,0 a 12,5 a 20,91 a 13,0 a 22,41 a 14,0 | | 4,45879 ± 0,173 | , 2,16910 ± 0,0100 | 2,000000 ± 0,181 2,17104 ±0,0512 | V |
| | 14,0 23,95 a 15,0 25,10 a 16,0 | | 4,115400 ± 0,187 3,887287 ± 0,228 | 1,40600 ±0,1245 2,70210 ± 0,172 2,70200 ± 0,132 2,504007 | 4,387180 ± 0,300 1,758828 ±0,880 | |
| | | | B,6078 ± 0,0071 | | 1,0000 4 0,00 | |
| | 20,0 33,05 22,0 34,05 22,5 34,65 23,0 35,95 24,0 | | 2,4460 ± 0,0162 2,1460 ± 0,0286 2,192908 ± 0,043 | 2,755107 ± 0,0211 2,1311000 ± 0,310 | 2,21 8300 ± 0,0088 2,161103, ± 0,0153 | |
| | 21,0 35,95 24,0 38,72 26,0 39,68 | | 2,391420 ± 0,220 2,27879 ± 0,0719 2,71220 ± 0,206 | | | |
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| Metallic compounds in the range of | 21568 S/020/61/ B103/B208 | /137/003/018/030 |
| Legend to Fig. 1: ordinate - | | |
| Hall constant v.cm; oe.a 10; o cast, annealed, sintered. | | |
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\$/062/61/000/002/001/012 B115/B207

18 7520 AUTHORS:

Card 1/8

1045, 1454, 1418

Kornilov, I. I. and Pylayeva. Ye. N.

TITLE:

Equilibrium of the ternary system of Ni₃Nb - Ni₃Ti - Ni₃Ta metallides

Izvestiya Akademii nauk SSSR. Otdeleniye khimicheskikh PERIODICAL:

nauk, no. 2, 1961, 197-200

TEXT: The authors studied the equilibrium of the ternary system of the metallic compounds Ni3Nb - Ni3Ti - Ni3Ta by physicochemical analysis (Fig. 1). In a previous study of binary systems consisting of Ni3Nb - Ni3Ta, Ni3Nb - Ni3Ti, Ni3Ta - Ni3Ti, the authors proved the existence of continuous, solid solutions by way of physicochemical analysis. On the basis of X-ray analysis, they determined the isostructural character of the compounds Ni3Nb and Ni3Ta. They are ascribed to the rhombic syngony of the structural type β - Cu_3Ti , and form continuous, solid solutions between each other.

S/062/61/000/002/001/012 B115/B207

Equilibrium of the ternary system ...

solubility of Ni3Ti - Ni3Ta - Ni3Nb indicates, for the compound Ni3Ti, the possible existence of a second high-temperature modification of the β - Cu_3Ti type rhombic syngony, as it is the case with the compounds The authors stress that no published data exist on Ni₃Nb and Ni₃Ta. the equilibrium in the mentioned ternary system, apart from a brief mention of the possibility of formation of continuous, solid solutions. Fig. 2 shows the composition of the alloys studied. A table provides data on the thermal analysis and the stability of alloys of the ternary system. The authors plotted the liquidus surface of the ternary system on the basis of thermal analysis data of three binary systems formed by the compounds, and of the three polythermal cross sections of the ternary system. The liquidus surface consists of a field of primary crystallization of continuous, ternary solid metallide solution of the system Ni₃Nb + Ni₃Ta + Ni₃Ti. Microstructural analyses of cast and annealed alloys confirm the existence of solid solutions in the ternary system. Fig. 6a shows the cast (mostly dendritic) structure of the

Card 2/8

S/062/61/000/002/001/012 B115/B207

Equilibrium of the ternary system ...

15% Ni, Nb, 15% Ni, Ta, and 70% Ni, Ti. Fig. 6b - of the alloy: 17% Ni, Nb, 33% Ni, Ti, 50% Ni, Ta, and Fig. 6c - of the alloy: 12% Ni3Nb, 70% Ni3Ta, 18% Ni3Ti. The microstructure of alloys of the same composition has become polyhedral after annealing at 1200°C for 24 hr (Figs. 6d, e, f). Finally, the authors studied the hardness in

the cast and the annealed state. The table shows the results of measurements of polythermal cross sections. Not only microstructure, but also hardness confirm the data of thermal analysis on the existence of continuous, solid solutions of metallides in the ternary system.

There are 7 figures, 1 table, and 7 Soviet-bloc references.

ASSOCIATION:

Institut metallurgii im. A. A. Baykova Akademii nauk SSSR

(Institute of Metallurgy imeni A. A. Baykov, Academy of

Sciences USSR)

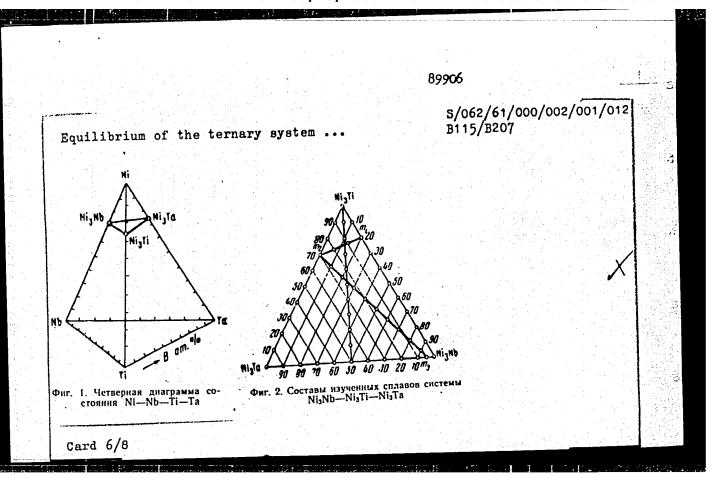
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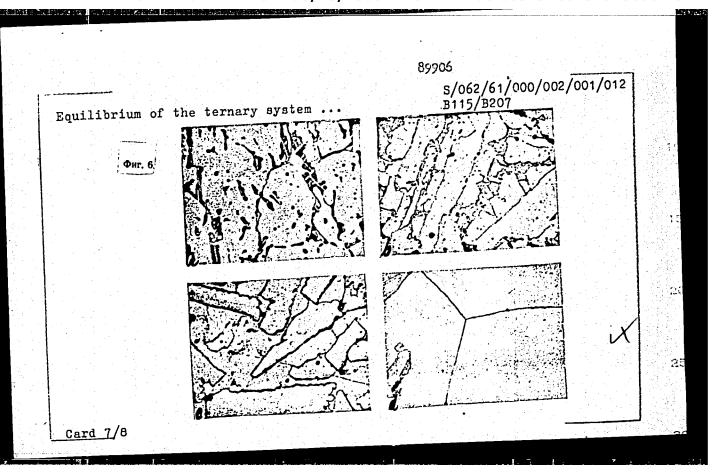
October 2, 1959

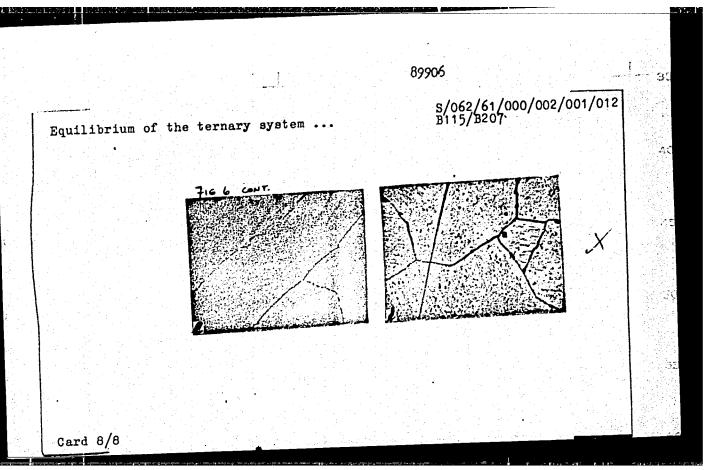
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| Legend to the to | able: 1) Composition ion temperature, °C, 3 | according to synthesis, % by weig) hardness H _V , kg/mm ² , 4) cast ation of cross sections, | At, |







PYLAYEVA, Ye.N.

18(2)

PHASE II - ABSTRACTS

AB-1

Akademiya nauk SSSR. Institut metallurgii

Titan i yego splavy; metallurgiya i metallovedeniye (Titanium and Its Alloys; Metallurgy and Physical Metallurgy) Moscow, Izd-vo AN SSSR, 1958. 209 p. 4,000 copies printed.

Resp. Ed.: N.V. Ageyev, Corresponding Member, USSR Academy of Sciences; Ed. of Publishing House: V.S. Rzheznikov; Tech. Ed.: A.A. Kiseleva.

INTRODUCTION: This book, of which a Phase I Exploitation (SOV/1200) has been prepared, is a collection of scientific papers devoted to the study of titanium and its alloys from three main points of view: physical metallurgy, forming, and welding. Special problems inphysical metallurgy, forming, and welding. Special problems investigated include structural changes occurring during welding, development of the content of harmful gases, development of industrial methods of rolling, and oxidation at various temperatures.

PART I. PHYSICAL METALLURGY

Ageyev, N.V., and L.A. Petrova (Institute of Metallurgy, USSR Academy of Sciences). Stability of the Beta Phase in Titanium-Molybdenum 3 Alloys Card 1/43

Titanium and Its Alloys (Cont.)

AB-1

decrease in lattice parameter. 4) Formation of the omega phase during the decomposition of the beta phase causes an increase in hardness in the investigated alloys, and is also the cause of brittleness observed in alloys containing 5.42-6.93 percent of Mn after heating in the 500-200° range, with holding times of 6-16 hours. Precipitation of the alpha phase is accompanied by a drop in hardness. There are 8 figures, 2 tables, and 5 references (1 Soviet and 4 English).

Kornilov, I.I., P.B. Budberg, M.A. Volkova, V.F. Prokhanov, Ye.N. Pylayeva (Institute of Metallurgy, USSR Academy of Sciences) Development of a Method for the Hot Compaction of Titanium and Titanium-Alloy Powders

The purpose of this investigation was to develop a satisfactory method of hot-compacting titanium powder. The authors first attempted hot compaction with graphite compression molds, which, however, proved unsatisfactory because the titanium reacts with the graphite and the molds can be used only once. The authors therefore used a new complex nickel alloy [composition not given] developed at the Institute of Metallurgy at the USSR Academy of Scineces in 1953-54. This alloy is some 40-50 times stronger than pure Ti at 950-1000° C. The alloy can therefore be recommended as Card 7/43

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- Titanium and Its Alloys (Cont.)

AB-1

a material for compression molds for hot compaction of powdered Ti, Be, Zr, Ni, Fe, Th, U, and other metals. Compression molds of the new alloy were made in the following shapes and sizes: 1) cylindrical, with 15-mm diameter, 20-mm height, and 15-g. weight; (2) cylindrical, with 45-mm.diameter, 60-mm, height, and approximately 400-g weight; (3) rectangular, 6x6x60 mm, 10 g.in weight. These molds were designed by one of the authors (V.F. Prok-A study was made of the effect of temperature, specific pressure, and duration of hot compaction on the density and hardness of the compact. Hot compaction of CaH2-reduced and Mg-reduced Ti was carried out at 800°, 850°, and 900° C, at a specific pressure of 15 kg/mm?, and for periods of 0.5 to 30 minutes. An investigation was also made of the hot compaction of Ti alloys containing 5 percent and 7.5 percent of Al. These tests were carried out at a temperature of 850° and at a specific pressure of 15 kg/mm² after preliminary sintering at 1000°. Conclusions. 1) The new heat-resistant nickel alloy may be used for making compression molds intended for hot compaction of metal powders at temperatures of 800-1000° C and at a specific pressure of 12-15 kg/mm. 2) It was established that the theoretical density of Ti is achieved by hot compaction Card 8/43

Titanium and Its Alloys (Cont.)

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with a specific pressure of 15 kg/mm² at 900° after 10 minutes, at 850° after 20 minutes, and at 800° after 30 minutes. 3) In the case of powdered titanium-aluminum alloys containing 5 percent and 7.5 percent of aluminum, hot compaction at 850° with a specific pressure of 15 kg/mm² for a period of 20 minutes is sufficient to obtain a density equal to 98 percent of the theoretical density of the alloys. 4) The proposed method of hot compaction may be used for other powdered metals (Zr, Be, Th, U, Fe, etc.) and for their alloys. There are 5 figures, 3 tables, and 10 references (8 English and 2 German).

Savitskiy, Ye.M., M.A. Tylkina, A.N. Turanskaya (Institute of Metallurgy, USSR Academy of Sciences) Recrystallization Diagrams of Titanium and Its Alloys

The aim of this investigation, conducted in 1954-55, was to study the process of recrystallization of titanium of various degrees of purity and of its alloys under conditions of various types of deformation and to construct two types of three-dimensional diagrams of the recrystallization process. Type I diagrams show the relationship between grain size, the degree of cold working, and the temperature of subsequent annealing, and can be used in establishing correct conditions for the annealing of semifinished Card 9/43

78-3-3-22/47 Kornilov, I. I., Pylayeva, Ye. N. AUTHORS: Investigations of the Binary Systems Ni 3Ti -Ni 3Ta and Ni 3Ti-Ni3Nb (Issledovaniye dvoynykh sistem Ni3Ti-Ni3Ta i Ni3Ti-TITLE : -Ni3Nb) The Binary System Ni3Ti Ni3Ta (Dvoynaya sistema Ni 3Ti-Ni 3Ta) Zhurnal Neorganicheskoy Khimii, 1958, Vol. 3, Nr 3, pp. 673-677 PERIODICAL: (USSR) In the present work the phase diagrams of the binary systems Ni Ti-Fi Ta and Ni Ti-Ni Nb were investigated. The phase ABSTRACT 8 diagrams of the binary systems between metallic compounds were determined by thermal analysis, microstructure analysis as well as investigations of the electric resistance, the hardness and the specific weight. On the basis of these investigations the phase diagrams were constructed. The compound Ni Ti crystallizes at 1375 C and the compound Ni Ta at 1531 C. The temperature of the crystallization of the alloys in the Card 1/2

Investigations of the Binary Systems Ni3Ti-Ni3Ta and Ni3Ti-Ni3Nb. The Binary System Ni3Ti-Ni3Ta

system Ni₃Ti-Ni₃Ta is lower than in pure compounds. The fusion diagram in the system Ni₃Ti-Ni₃Ta represents an uninterrupted series of solid solutions between the compounds and the minimum crystallization temperature lies at 30 % Ni₃Ta. The microstructure of the alloy in the state of equilibrium (after 200 hours treatment at 1200°C) shows polyhedral crystals. The fusion diagram of the system Ni₃Nb-Ni₃Ti is based on the thermal analysis, the determination of the microstructure, the hardness, the electric resistance and the specific weight of the alloys. The melting point of the compound Ni₃Nb lies at 140°C. By addition of Ni₃Ti to the compound Ni₃Nb at 70 % Ni₃Ti the minimum of the melting point is 1285°C. There are 3 figures, 2 tables, and 11 references, 9 of which are Soviete

ASSOCIATION:

Institut metallurgii im. A. A. Baykova, Akademii nauk SSSR (Metallurgical Institute imeni A. A. Baykov, AS USSR)

SUBMITTED:

June 25, 1957

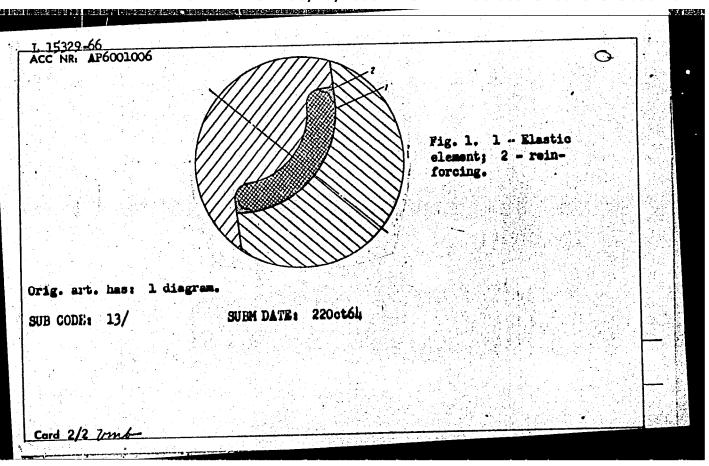
Card 2/2

PYLAYEVA, Te.N.; GLADYSHEVSKIY, Te.I.; KRIPTAKEVICH, P.I.

Crystalline structure of NigNb and NigTa compounds. Zhur.
neorg. khim. 3 no.7:1626-1631 J1 '58. (MIRA 11:9)

1. Institut metallurgii im. A.A.Baykova AN SSSR i L'vovskiy
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(Nickel niobide) (Nickel tantalide)

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ACCESSION NR: AP4011134

5/0182/64/000/001/0021/0024

AUTHORS: Grinshpun, L. Ya.; Pywlaykin, P. A.; Khirdzhiyev, Ye. V.; Pertsovskaya, Yo. V.

TITLE: Containers of high power horizontal hydraulic presses for pressing aluminum alloys

SOURCE: Kuznechno-shtampovochnoye proizvodstvo, no. 1, 1964, 21-24

TOPIC TAGS: hydraulic press, press container, 5KhNV steel, 5KhNM steel, 5KhNM2 steel, 38Kh2N3M steel, 3Kh2N2MVF steel, 27Kh2N2MVF steel

ABSTRACT: The technological requirements of containers for pressing Al alloys were limited by the temperatures up to 430C, specific stresses up to 50 kg/mm², and the maximum press force 12 000 T. A commonly used container consisted of a frame and a conical bushing. Both the frame and the bushing were made of high-alloy steels 5KhNV or 5KhNH. They had a number of shortcomings associated with the shape of the bushing and the metal used. For this reason, several research projects leading to the design of more suitable containers were undertaken at

Cord 1/3.

ACCESSION NR: APholl134

the Uralmashzavod (Ural Machine Plant). The new types had multilayer frames and cylindrical bushings (see Fig. 1 of the Enclosure). The problem of obtaining steels with high mechanical properties (5 > 150 kg/mm²) at 4800 has not yet been solved. The steels studied so far were: 5KhNM2, 38Kh2N3M, 3Kh2N2HVF and 27Kh2N2HVF. A standard mathematical procedure for calculating the strength of a multilayered thick-wall cylinder subjected to internal pressure is presented. Orig. art. has: 1 table, 3 figures, and 2 formulas.

ASSOCIATION: none

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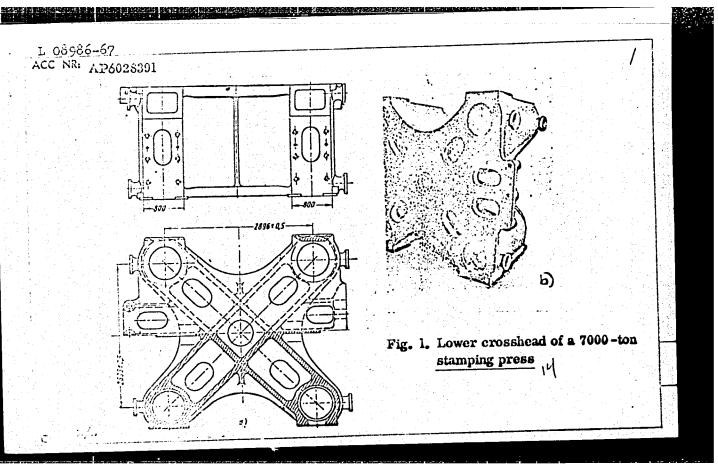
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| ACT.OR: Pylaykin, P. A. | |
| Olice none) | |
| TTTLE: A new design solution for the crosshead of a four-column press | |
| SOURCE: Kuznechno-shtampovochnoye proizvodstvo, no. 6, 1966, 26-27 | |
| TOPIC TAGS: forge press, metal forming press, box beam, foundry equipment | |
| ABSTRACT: The Uralmashzavod Machinery Plant has recently developed a new design of crossheads for four-column forging and stamping presses with capacities of up to 7000 tons, as exemplified by the lower crosshead of a 7000-ton stamping press shown in Fig. 1. This design displays the following advantages over the conventional press crossheads: it is formed by two crossed diagonal box beams, and it assures a statically determinate energy diagram. The walls of the beams lack the apertures customarily required for fixing and centering the mold cores and treating the blowholes; such apertures sharply weaken the loadbearing strength of the design; in the new design the side walls of the beams have been left open for this purpose. In addition, the box-like cross section of the crosshead enables it to withstand the torque that | |
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ACC NR: AP7003551

SOURCE CODE: UR/0023/66/000/004/0519/0530

AUTHOR: Pyldmaa, V.

ORG: Institute of Physics and Astronomy, Academy of Sciences, Estonian SSR (Institut fiziki i astronomii Akademii nauk Estonskoy SSR)

TITLE: The importance of multiple scattering in twilight

SOURCE: AN EstSSR. Izvestiya. Seriya fiziko-matematicheskikh i tekhnicheskikh nauk, no. 4, 1966, 519-530

TOPIC TAGS: twilight, light scattering, earth atmosphere, sky brightness, journ National North Report of the National North Report of the National National North Report of the National National North Report of the National Natio

ABSTRACT: The effect of high-order scattering on the brightness of the twilight sky is described. The brightness of the sky in absolute units of energy was measured by the author in 1963 in three narrow spectral ranges (λ = 422; 479; 574 mµ). The dependence of total sky brightness on the Sun's vertical is found for three directions: in the zenith by G. Rosenberg's method, and in directions $z=\pm70^{\circ}$ by V. Fesenkov's precisely defined method. The changes in sky brightness caused by the first (I_1), and by higher-order (I_2) scattering as a function of the Sun's zenithal distance ℓ are examined. The peculiarities of the change of the relation I_2/I_1 in different phases of twilight are analysed. The brightnesses I_1 and I_2 in different regions of the spectrum for different zenith distances of the Sun are compared. The dependence of the color index on ℓ is analysed. It is also shown that changes in the spectral composition of diffuse radiation of the twilight sky and I/2

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PYLLMAA, V. [Põldmaa, V.]

Spectrophotometer for airglow observation. Izv. AN Est. SSR. Ser. fiz.-mat. i tekh. nauk 13 no.3:192-199 '64.

(MIRA 17:11)

1. Academy of Sciences of the Estonian SSR, Institute of Physics and Astronomy.

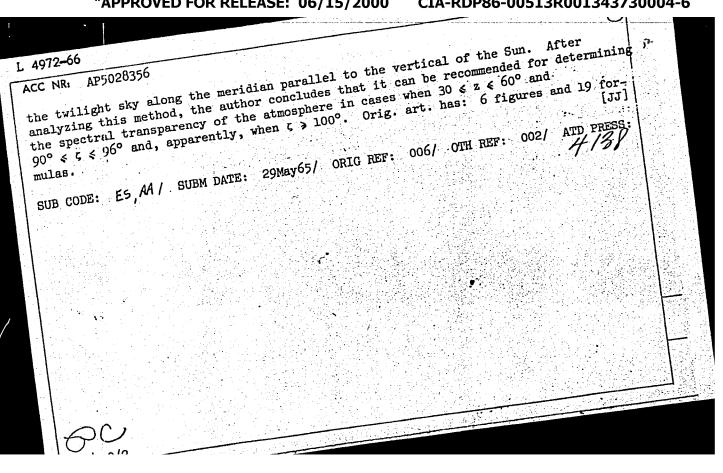
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Interpretation of some characteristics of the brightness picture of a twilight sky. Izv. AN SSSR. Fiz. atm. i ckeana 1 no.11:1168-1177 N 65. (MIRA 18:12)

1. Institut fiziki i astronomii AN Estonskoy SSR. Submitted May 29, 1965.

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